

UNREPRODUCTION COPY
SUBJECT TO REVISION
IN TWO WEEKS

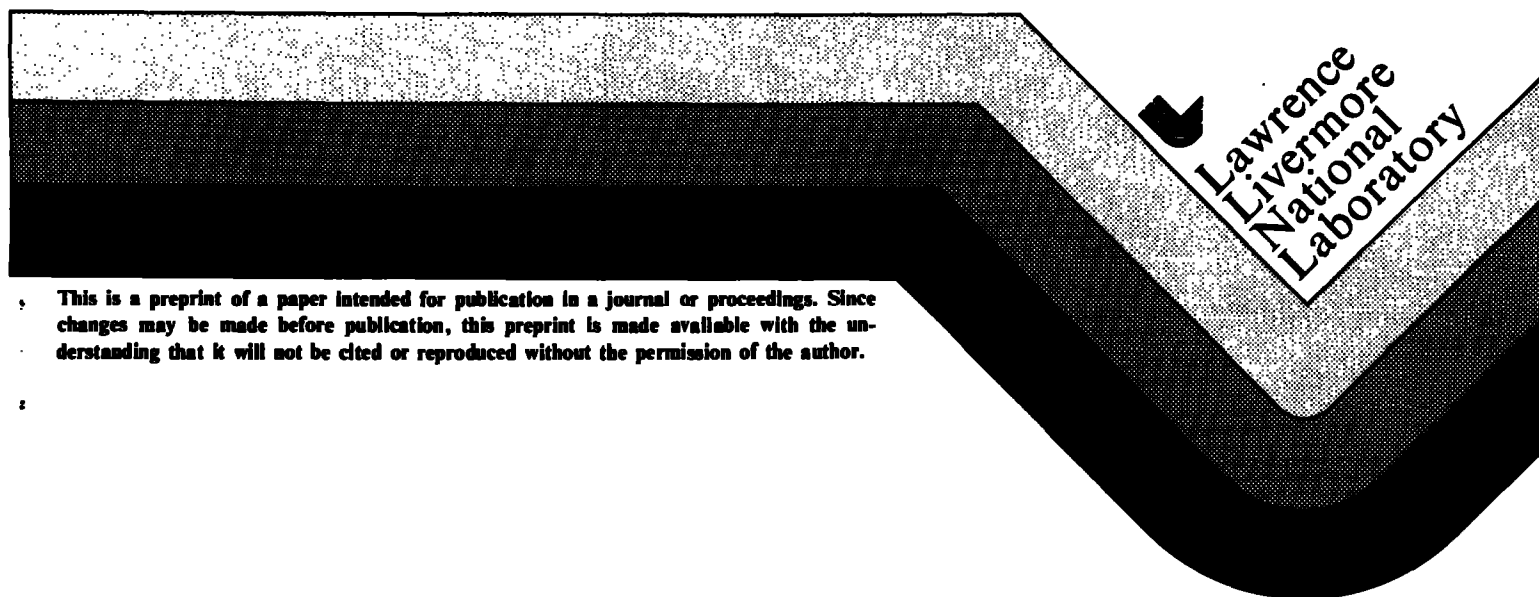
UCRL- 94476
PREPRINT

THE Be-Si (BERYLLIUM-SILICON) SYSTEM

H. Okamoto
L. E. Tanner

This paper was prepared for submittal to
Bulletin of Alloy Phase Diagrams

April 18, 1986



This is a preprint of a paper intended for publication in a journal or proceedings. Since changes may be made before publication, this preprint is made available with the understanding that it will not be cited or reproduced without the permission of the author.

DISCLAIMER

This document was prepared as an account of work sponsored by an agency of the United States Government. Neither the United States Government nor the University of California nor any of their employees, makes any warranty, express or implied, or assumes any legal liability or responsibility for the accuracy, completeness, or usefulness of any information, apparatus, product, or process disclosed, or represents that its use would not infringe privately owned rights. Reference herein to any specific commercial product, process, or service by trade name, trademark, manufacturer, or otherwise, does not necessarily constitute or imply its endorsement, recommendation, or favoring by the United States Government or the University of California. The views and opinions of authors expressed herein do not necessarily state or reflect those of the United States Government or the University of California, and shall not be used for advertising or product endorsement purposes.

The Be-Si (Beryllium-Silicon) System

9.01218 28.0855

By H. Okamoto and L.E. Tanner

Lawrence Livermore National Laboratory

Equilibrium Diagram

The assessed Be-Si phase diagram (Fig. 1) is based on the experimental work of [29Mas], who used only four specimens. The observed liquidus data do show a degree of uncertainty (see also [Hansen]). [46Ray] expected intermetallic compound formation based on the electronegativity of Si; however, the absence of intermediate phases has been confirmed by several investigators (e.g., [32Slo], [50Kaul], [61Hin], [62Jor]). The solubility ranges of the terminal phases appear to be quite narrow.

(β Be) and (α Be) Terminal Solid Solutions. The melting point of β Be and the β Be \rightarrow α Be allotropic transformation temperature are 1289 ± 4 and 1270 ± 6 °C, respectively [85BAP]. Metallographic [32Slo, 50Kaul] and X-ray lattice parameter [50Kau, 62Jor] studies did not reveal a solubility range for the (α Be) phase. [61Hin] observed that an alloy of 0.31 at.% Si was two-phase. Therefore, the solubility limit must be at least less than this value. Thermal analysis was not successful in defining the solubility limit of the (β Be) [61Gel].

(Si) Terminal Solid Solution. The melting point of Si is 1414 °C [Melt]. Solubility of Be in Si could not be detected by X-ray lattice parameter measurements [62Jor]. More recently, [82Tom] estimated the solid solubility of electrically active Be in (Si) from the resistivity profile of a diffusion couple. Solubility was expressed as $2.2 \times 10^{20} \exp(-1.97 \times 10^4/T)$ atoms/cm³ for the temperature range between 800 and 1000 °C (or $0.402 \exp(-1.97 \times 10^4/T)$ at.% assuming that the density of Si is 2.55 g/cm³).

The lattice parameter changes in various alloys suggested the existence of some sizable solubility of Be in (Si) [61Jor1] or Si in (α Be) [61Jor2]. However, further detailed studies on rapidly quenched alloys did not reveal any solubility [62Jor].

L \rightarrow (α Be) + (Si) Eutectic Reaction. The liquidus and eutectic temperatures were measured by [29Mas].

Composition, at.% Si	Liquidus, °C	Eutectic, °C
0	1278	
7.4	1205	1065
24.3	1142	1090
42.8	1119	1090
64.5	1255	1088
100	1407	

The eutectic composition was proposed to be 33 at.% Si from the trend of arrest time with Si concentration at the eutectic temperature [29Mas]. [48Haj] redetermined the eutectic composition to be 38.5 at.% Si, presumably by reevaluating the liquidus trend due to [29Mas]. A best estimate based on the limited data is that the eutectic falls within the range 36 ± 3 at.% Si.

Crystal Structures

A summary of crystal structure and lattice parameter data for pure elements is given in Table 1.

Thermodynamics

No experimental thermodynamic data are available. [78Ball] estimated the interaction parameter of liquid Be-Si by considering electronic and internal pressure effects. According to their regular solution and quasi-chemical models (Table 2), the calculated eutectic composition is displaced significantly toward Be-rich compositions (Fig. 2). The lattice stability parameters of Be and Si have been assumed (Table 2) from the heats of transformation given by [83Chal]. These may differ from the values used (but not given) by [78Ball]. The calculated diagram presented here is based on a subregular solution model and agrees slightly better with the experimental liquidus data. Further improvement of the modeling may be attained when the liquidus boundaries are better established through the use of purer materials and modern techniques.

[77Sud] calculated the activity of Si from the phase diagram given by [Hansen].

Table 1 Be-Si Crystal Structure and Lattice Parameter Data

Phase	Composition, at.% Si	Struktur- Pearson bericht symbol designation		Space group	Proto- type	Lattice parameters, nm		Reference
						a	c	
(β Be)....	0	cI2	A2	Im3m	W	0.25515	...	[King2]
(α Be)....	0	hP2	A3	P6 ₃ /mmc	Mg	0.22857	0.35839	[King1]
(Si).....	100	cF8	A4	Fd3m	diamond	0.54306	...	[King1]

Table 2 Thermodynamic Functions for the Be-Si System (J/mol)

Lattice Stability Parameters:

$$\begin{aligned}
 G^\circ(\text{Be}, \text{L}) &= 0 \\
 G^\circ(\text{Si}, \text{L}) &= 0 \\
 G^\circ(\text{Be}, \text{bcc}) &= -12600 + 8.067T \\
 G^\circ(\text{Be}, \text{cph}) &= -14700 + 9.428T \\
 G^\circ(\text{Si}, \text{diamond}) &= -50210 + 29.763T
 \end{aligned}$$

Excess Gibbs Energy of Mixing for the Liquid Phase:

$$\begin{aligned}
 G^{\text{ex}} &= 14916X(1-X) && \text{[78Ba1] (Regular Solution Model)} \\
 G^{\text{ex}} &= 14916X(1-X)(1 - 449X(1-X)/T) && \text{[78Ba1] (Quasi-Chemical Model)} \\
 G^{\text{ex}} &= X(1-X)(19142 - 20027X) && \text{[Present] (Subregular Solution Model)}
 \end{aligned}$$

X: mole fraction of Si. T: Temperature in K.

Acknowledgements

Be-Si evaluation contributed by L.E. Tanner, L-217, Lawrence Livermore National Laboratory, P.O. Box 808, Livermore, CA 94550 and H. Okamoto, B77G, Lawrence Berkeley National Laboratory, Berkeley, CA 94720. Work was supported by the U.S. Department of Energy under contract no. W-7405-Eng-48 and American Society for Metals (ASM). Literature searched through 1984. Part of the bibliographic search was provided by ASM. L.E. Tanner and H. Okamoto are ASM/NBS Data Program Category Editors for binary beryllium alloys.

General References

- Hansen: M. Hansen and K. Anderko, Constitution of Binary Alloys, McGraw-Hill, New York or General Electric Co., Business Growth Services, Schenectady, NY 12345 (1958)
- Elliott: R.P. Elliott, Constitution of Binary Alloys, First Supplement, McGraw-Hill, New York or General Electric Co., Business Growth Services, Schenectady, NY 12345 (1965)
- Shunk: F.A. Shunk, Constitution of Binary Alloys, Second Supplement, McGraw-Hill, New York or General Electric Co., Business Growth Services, Schenectady, NY 12345 (1969)
- Pearson: W.B. Pearson, Handbook of Lattice Spacings and Structures of Metals and Alloys, Vol. 1 (1958) and Vol. 2 (1967), Pergamon Press, New York
- Hultgren,B: R. Hultgren, P.D. Desai, D.T. Hawkins, M. Gleiser, K.K. Kelley, and D.D. Wagman, Selected Values of the Thermodynamic Properties of the Elements, American Society for Metals, Metals Park, OH (1973)
- Hultgren,B: R. Hultgren, P.D. Desai, D.T. Hawkins, M. Gleiser, and K.K. Kelley, Selected Values of the Thermodynamic Properties of Binary Alloys, American Society for Metals, Metals Park, OH (1973)
- Melt: "Melting Points of the Elements", Bull. Alloy Phase Diagrams, 2(1), 145-146 (1981)
- King1: H.W. King, "Crystal Structures of the Elements at 25 °C, Bull. Alloy Phase Diagrams, 2(3), 401-402 (1981)
- King2: H.W. King, "Temperature-Dependent Allotropic Structures of the Elements", Bull. Alloy Phase Diagrams, 3(2), 275-276 (1982)

Cited References

- *29Mas: G. Masing and O. Dahl, "Aluminum Alloys Containing Beryllium, Silicon-Beryllium Alloys," Wiss. Veroffentl. Siemens-Konzern, 8(1), 248-256 (1929) in German. (Equi Diagram; Experimental; #)
- 32Slo: H.A. Sloman, "Researches on Beryllium," J. Inst. Metals, 49, 365-388 (1932). (Equi Diagram; Experimental)
- 46Ray: G.V. Raynor, "Beryllium, Beryllium Alloys, and the Theoretical Principles Affecting Alloy Formation with Beryllium," J. Roy. Aeronaut. Soc., 50, 410 (1946). (Equi Diagram; Theory)
- 48Haj: O. Hajicek, "Binary Alloys Containing a Eutectic," Hutnicke Listy, 3, 265-270 (1948) in Czeck. (Equi Diagram; Compilation)
- 50Kau: A.R. Kaufmann, P. Gordon, and D.W. Lillie, "The Metallurgy of Beryllium," Trans. ASM, 42, 785-844 (1950). (Equi Diagram; Experimental)
- 61Gel: S.H. Gelles, J.J. Pickett, E.D. Levine, and W.B. Nowak, "The Stability of the High-Temperature Phase in Beryllium and Beryllium Alloys," Institute of Metals Conference on Metallurgy on Beryllium, London, Preprint no.33, 11pp (1961); Inst. Metals, Monograph Rept. Ser. No. 28, The Metallurgy of Beryllium, 588-600, Institute of Metals, London (1963). (Equi Diagram; Experimental)
- 61Jor1: C.B. Jordan, "Investigation of the Effect of Ultrarapid Quenching on Metallic Systems, Including Beryllium Alloys," Monthly Prog. Rept. AD 270977, 3pp (Aug. 1961). (Equi Diagram; Experimental)
- 61Jor2: C.B. Jordan, "Investigation of the Effect of Ultrarapid Quenching on Metallic Systems, Including Beryllium Alloys," Monthly Prog. Rept. AD 268088, 4pp (Nov. 1961). (Equi Diagram; Experimental)
- 61Hin: E.D. Hindle and G.F. Slattery, "A Metallographic Survey of Some Dilute Beryllium Alloys," Institute of Metals Conference on Metallurgy of Beryllium, London, Preprint no. 58, 8 pp. (1961); Inst. Metals, Monograph Rept. Ser. No. 28, The Metallurgy of Beryllium, 651-664, Institute of Metals, London (1963). (Equi Diagram; Experimental)
- 62Jor: C.B. Jordan, "Investigation of the Effect of Ultrarapid Quenching on Metallic Systems, Including Beryllium Alloys," Tech. Rept. ASD-TDR-62-181, AD 284409, 33pp (1962). (Equi Diagram; Experimental)
- 77Sud: V.S. Sudavstove and G.I. Batalin, "Calculation of the Components of Molten Metal-Si Alloys from the Phase Diagrams," Ukr. Khim. Zh. 43(3), 235-240 (1977) in Russian. (Thermo; Theory)
- 78Bal: S.S. Balakrishna and A.K. Mallik, "Synthesis of Binary Metallic Systems," Applications of Phase Diagrams in Metallurgy and Ceramics. Vol. 2, NBS Special Pub. 496, 1200-1219 (1978). (Thermo; Theory; #)
- 82Tom: H. Tomokage, M. Hagiwara, and K. Hashimoto, "Diffusion Coefficient and Solid Solubility of Electrically Active Beryllium in Silicon," Mem. Fac. Eng., Kyushu Univ., 42(2), 89-94 (1982). (Equi Diagram; Experimental)
- 83Cha: M.W. Chase, "Heats of Transformation of the Elements," Bull. Alloy Phase Diagrams, 4(1), 123-124 (1983). (Thermo; Compilation)
- 85BAP: to be published in Bull. Alloy Phase Diagrams (1985). (Equi Diagram; Compilation)

* Indicates key paper.

Indicates presence of a phase diagram.

Be-Si Figure Captions

Fig. 1 Be-Si Assessed Phase Diagram

H. Okamoto and L.E. Tanner, 1985

Fig. 2 Comparison of Be-Si Assessed and Calculated Phase Boundaries

Thermodynamic models are given in Table 2.

———— = Assessed, ————— = Present model (Subregular),
———— = [78Ba1] (Regular), ————— = [78Ba1] (Quasi-chemical).

H. Okamoto and L.E. Tanner, 1985

